**CONTACT MASS IN THE DIRECT SYNTHESIS OF ALKOXYSILANES. RAMAN SPECTROSCOPY STUDY** 

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#### **Abstract**

Raman spectroscopy was used to study the composition of contact masses (CMs) CuCl + Si obtained by different methods, which are utilized in the direct synthesis of alkoxysilanes. The CMs prepared by the classical method were shown to include copper oxides in a mixture with silicon and on the surface of metallic copper and  $Cu<sub>5</sub>Si$  intermetallide. Along with copper oxides, an active phase containing terminal Cu–Cl groups was also observed. In the case of mechanochemical synthesis, the formation of copper oxides was not observed. In view of the metallic conductivity of  $Cu<sub>3</sub>Si$  and  $Cu<sub>5</sub>Si$ , their direct identification using Raman spectra is not possible. All the mentioned copper-containing phases are involved in the reaction with alcohols and esters as catalysts. A difference between the two methods for obtaining CMs comes down to the presence or absence of a "catalytic cocktail", which significantly affects the parameters of the direct synthesis course.



**Key words:** direct synthesis of silicones, contact mass, copper chloride, silicon, Raman scattering.

# **Introduction**

Nowadays, there is a revival of interest in the direct synthesis of alkoxysilanes and alkyl alkoxysilanes by the interaction of silicon with alcohols, esters, or alkenes [1–15]. Despite the significant progress in the implementation of the classical chloride synthesis through the Rochow reaction (see review [16]), the rising environmental concerns lead to tightening of the requirements for reducing harmful emissions. In this respect, the direct synthesis of alkoxysilanes offers a more environmentally benign route to the production of silicones [17], which is in good agreement with the main principles of green chemistry [18].

Due to the high inertness of Si and organic reagents, the direct synthesis requires the addition of metal catalysts which are usually based on copper [1–7]. The first step of the synthesis includes the production of the so-called contact mass, in the classic version, a mixture of Si and CuCl heated to 200–350 °C in an inert atmosphere. This results in the formation of metallic copper and  $Cu<sub>x</sub>Si$  intermetallides according to the following reactions [19].

> $7Si + 12CuCl \rightarrow 3SiCl_4 + 4Cu_3Si$  $9Cu<sub>5</sub>Si + 8CuCl \rightarrow 2SiCl<sub>4</sub>+ 7Cu<sub>5</sub>Si$  $Cu<sub>5</sub>Si + 4CuCl \rightarrow SiCl<sub>4</sub>+ Cu$

It is copper silicides, in particular,  $Cu<sub>3</sub>Si$  that are believed to comprise an active catalytic phase which interacts with alcohols or esters [9–15]. Furthermore, the reaction is assumed to

proceed through the formation of reactive silylenes on the surface [20, 21].

The phase composition of the contact mass was commonly analyzed by powder X-ray diffraction (XRD), and the results obtained in all the cited works confirm the formation of  $Cu<sub>3</sub>Si$  or  $Cu<sub>5</sub>Si$  intermetallide. The goal of this work was to create the basis for controlling the process of the direct synthesis at different stages using Raman microspectroscopy. Unlike other methods, the Raman spectra can be registered *in situ* and offer the advantage of spatial resolution.

## **Experimental section**

The work was concerned with the following commercial reagents: CuO (99%, ABCR AB180315), Cu<sub>2</sub>O (99%, ABCR AB209068), CuCl (97%, Sigma-Aldrich 212946), Cu<sub>5</sub>Si (Alfa Aesar 12844), metallic copper particles 3 μm in size (Alfa Aesar 00622), and technical silicon (KR-1, purity >98%, impurities: Fe  $\leq 0.7\%$ , Al  $\leq 0.7$ , Ca  $\leq 0.6\%$ ).

All the conditions and methods for preparing the CMs are presented in Table 1. To prepare the contact mass by the classical method (**CM1**), a mixture of technical silicon (Si content 98%, particle sizes  $\sim 60-100 \text{ }\mu\text{m}$ ) and CuCl in the mass ratio of 5:1 was heated to 350 °C for 30 min in an argon atmosphere. The composition and heating conditions were optimized to achieve the highest activity [14]. According to the elemental analysis data, sample **CM1** contained ~80 wt % of Si and 16 wt % of Cu. The second type of contact masses (**CM2**) containing 7.8 wt % of Cu and 92.2 wt % of Si was obtained by grinding Si and CuCl in a vibration ball mill at 250 °C for 2 h using steel balls [13, 15, 22]. Contact mass **CM3** containing 5.8 wt % of Cu and 91.2 wt % of Si was obtained by grinding Si and CuCl in a vibration ball mill at 250 °C for 3 h using brass balls. For the experiment purity, sample **CM4** was prepared by vacuum deposition of purified CuCl on a silicon wafer (Si 99.999%) for microelectronics treated with HF followed by heating to 350 °C for 30 min in an inert atmosphere. In addition, the precursors of contact masses **CM1** and **CM4** not subjected to heating (samples **CM1a** and **CM4a**) were studied.

**Table 1**. Methods and conditions for the synthesis of the CMs

<b>CM</b>	Si	Method for synthesis	T, °C	Time, min
CM1	Technical activated, 60- $100 \mu m$ fraction	Agitation of the powders in an argon atmosphere	350	30
CM2	Technical, 1.0- 1.5 mm fraction	Grinding in a vibration mill with steel balls	250	120
CM3	Technical, 1.0- 1.5 mm fraction	Grinding in a vibration mill with brass balls	250	180
CM4	Single crystalline for microelectro- nics, wafer	Surface activation with HF and vacuum deposition of purified CuCl	350	30

The presence of Cu3Si intermetallide in **CM1**, **CM2**, and **CM3** was confirmed by the results of X-ray diffraction analysis. The X-ray diffraction patterns in the range of  $20-100^{\circ}$  (2 $\theta$ ) were obtained using a Proto AXRD θ-2θ diffractometer equipped with an X-ray Cu tube with a kβ filter ( $\lambda = 1.541874$  Å) and a Detris Mythen 1K 1D detector.

The Raman spectra were recorded using a Horiba JobinYovin LabRam 300 Raman spectrometer equipped with a CCD detector and an Olympus BX2 microscope (objectives M-Plan 10x, 50x and 100x). The spectra were excited with a 632.8 nm He–Ne laser with a power of no more than 1 mW. During the investigation, ~600 spectra were recorded.

#### **Results and discussion**

Technical silicon (KR-1, purity >98%, impurities: Fe < 0.7%, Al  $\leq$  0.7, Ca  $\leq$  0.6%) and metallic silicon for microelectronics (99.999%) were used as the initial components. Their Raman spectra show an intense TO line with a frequency of 520 cm–<sup>1</sup> , which is slightly broadened in the case of technical Si (Fig. S1 in the Electronic supplementary information (ESI)) due to slight disordering of the crystal lattice.

Commercial copper(I) chloride (purity degree 97%), which is commonly used for the production of contact masses, contains an admixture of  $Cu<sub>4</sub>(OH)<sub>6</sub>Cl<sub>2</sub>$  [16]; therefore, its Raman spectrum shows additional low-intensity lines (Fig. 1a) that correspond to the impurity spectrum [23]. The sublimation in a vacuum of  $10^{-2}$  Torr at 240–250 °C affords pure CuCl, which Raman spectrum displays lines at 120, 159, and 197  $\text{cm}^{-1}$  on a broad shoulder of the Rayleigh line (Fig. 1b). The Raman spectra of the samples of commercial, purified, and pure CuCl obtained by the published method [24] coincide on the positions of the main lines (Fig. 1) but differ from the spectrum of

crystalline γ-modification CuCl, which features narrow lines at 210, 174  $\text{cm}^{-1}$  and a broad low-intensity band at 156  $\text{cm}^{-1}$  [25]. Taking into account the XRD data (Fig. S2 in the ESI) [13, 14], it can be concluded that commercial CuCl used in the direct synthesis exists in the form of an α-modification.



**Figure 1.** Raman spectra of commercial CuCl (*a*), purified CuCl (*b*), and a CuCl film deposited on the Si wafer (*c*).

In sample **CM4a**, the presence of a CuCl layer on the Si wafer is confirmed by the CuCl lines (Fig. 1c) and the appearance of the silicon TO mode at  $520 \text{ cm}^{-1}$ . According to the results of Raman micromapping, sample **CM1a** which was not subjected to heating, unlike **CM4a**, as expected, is heterogeneous. In particular, it contains the regions with a pure silicon phase (dark shiny particles) and a mixture of CuCl + Si (light particles) (Figs. S3–S5 in the ESI).

The elucidation of heated contact masses **CM1** and **CM4** showed some similarities. Firstly, the heated samples have a darker brown color compared to **CM1a**. Secondly, they do not contain the particles with CuCl. Micromapping of **CM1** also revealed the presence of pure silicon (Fig. 2a), which was expected in view of its initial excess, and the formation of new phases mixed with silicon and separately, such as metallic copper and phases with the Raman spectrum depicted in Fig. 2d. Mapping of more homogeneous sample **CM4** also revealed phase separation. In one of the regions, along with the visual presence of metallic copper, four different types of spectra were recorded (Fig. 3). Therewith, the Raman spectra of **CM1** and **CM4** particles presented in Figs. 2d and 3a, 2e and 3d coincide in pairs.

For the precise assignment of the spectra of the new phases, the spectra of commercial samples of finely dispersed  $(3 \mu m)$ copper,  $Cu<sub>2</sub>O$ ,  $CuO$ , and  $Cu<sub>5</sub>Si$  were recorded (Fig. 4). The Raman spectra of  $Cu<sub>2</sub>O$  (Fig. 4a) and CuO (Fig. 4b) coincide with the spectra described elsewhere [26] and contain the most intense narrow lines at 220 and 295  $cm^{-1}$  along with a broad overtone at  $~530 \text{ cm}^{-1}$ . Metallic copper does not have a vibrational spectrum; however, an oxide film is present on its surface, which was detected in the Raman spectra. Their analysis revealed two types of spectra (Fig. 4c,d) which differ by the presence of a narrow line at  $296 \text{ cm}^{-1}$  corresponding to CuO. A broad shoulder of the Rayleigh line, a narrow line at 146 and the broad one at  $~590 \text{ cm}^{-1}$  (Fig. 4c) are in good agreement with the Raman spectrum of the second modification of  $Cu<sub>2</sub>O$  [27].

When studying the surface of commercial intermetallide  $Cu<sub>5</sub>Si$ , two types of the spectra were detected (Fig. 4f,e) that differ by the presence and absence of CuO lines at 295 and 342  $\text{cm}^{-1}$ . The most characteristic band for Cu<sub>5</sub>Si is considered to be a broad band at 595  $cm^{-1}$ , which differs in the shape and lower frequency from the similar bands of copper oxides. Since intermetallic compound Cu<sub>5</sub>Si displays metallic conductivity, it does not have its own vibrational spectrum; therefore, the spectrum in Fig. 4f corresponds to the oxide film and can characterize it only indirectly.

The Raman spectra in Figs. 2e and 3d, recorded for **CM1** and **CM4** particles, exhibit a broad band at  $\sim$ 290 cm<sup>-1</sup> and lack the bands of copper oxides at 144, 220, 298, 343, and 600–630  $\text{cm}^{-1}$ . They can be assigned to an intermediate phase containing terminal Cu–Cl groups. The presence of surface Cu–Cl particles that prevent the diffusion of copper into silicon has been reported earlier [2].

It should be noted that Su *et al*. [28] mistakenly assigned the silicon TO line shifted to 513  $cm^{-1}$  due to the high power of the exciting laser [29] to Cu<sub>3</sub>Si phase. Sreedharan *et al.* [30] presented similar incorrect assignment.

The analysis of the morphology of **CM2** and **CM3** showed that the particles are aggregated, and their sizes do not exceed 5 µm (Fig. S6 in the ESI). **CM2** and **CM3** particles under consideration afforded the Raman spectra (several tens of the Raman spectra were recorded) similar to those depicted in Fig. 2a and correspond to crystalline silicon. Moreover, **CM2** and **CM3** did not contain the particles with the spectra similar to those presented in Figs. 2d,e and 3. Since according to the XRD data (Figs. S7–S9 in the ESI), these contact masses contain  $Cu<sub>3</sub>Si$  phase, it can be assumed that, due to metallic conductivity, intermetallic compound  $Cu<sub>3</sub>Si$  [31] cannot be identified using vibrational spectra. Apparently,  $Cu<sub>3</sub>Si$  does not form a detectable oxide film, unlike  $Cu<sub>5</sub>Si$ .

When analyzing the spectra, it was noted that the Raman spectra of the components such as CuCl, thin surface oxide films on Cu, and  $Cu<sub>5</sub>Si$  are very low intensive, and their registration is a laborious and time-consuming process. These peculiarities should be taken into account when monitoring *in situ* the direct synthesis using Raman spectroscopy.

In general, Raman spectroscopy studies of the contact masses showed that samples **CM1** and **CM4** have similar phase compositions and contain crystalline silicon, metallic copper, its oxides of three types, and an intermediate phase with terminal Cu–Cl bonds. The presence of  $Cu<sub>4</sub>(OH)<sub>6</sub>Cl<sub>2</sub>$  impurity in commercial CuCl, which provides a slight increase in the content of copper oxides in the contact mass, is not of fundamental importance. In contrast to samples **CM1** and **CM4**, no copper oxides were found in samples **CM2** and **CM3**. This is likely to be due to the fact that, under conditions of mechanical activation, all copper oxides are converted to  $Cu<sub>3</sub>Si$ . In all the samples explored, the presence of  $Cu<sub>3</sub>Si$  or  $Cu<sub>5</sub>Si$  cannot be directly confirmed by Raman spectroscopy due to their metallic conductivity.

The significantly more diverse phase compositions of contact masses **CM1** and **CM4**, in which different phases have different catalytic activity, are in good agreement with the



**Figure 2.** Raman micromapping of **CM1**.



**Figure 3.** Raman micromapping of **CM4**.



Figure 4. Raman spectra of Cu<sub>2</sub>O  $(a)$ , CuO  $(b)$ , and the oxides on the surface of metallic copper  $(c, d)$  compared to the spectrum of the oxide film on the surface of commercial intermetallide  $Cu<sub>5</sub>Si$  (*e*, *f*).

concept of a "catalytic cocktail" [32]. Mechanochemically prepared **CM2** and **CM3** do not contain a variety of catalysts like **CM1** and **CM4**, which explains the differences in their reactivity in the synthesis of alkoxysilanes [13–15, 22].

## **Conclusions**

The performed Raman microspectroscopy investigation of contact masses **CM1** and **CM4**, obtained by the classical method used in the direct synthesis of silanes, showed the presence of copper oxides of various modifications both on the surface of metallic copper and in a mixture with crystalline silicon, as well as the formation of the phase containing active terminal Cu–Cl groups. The contact masses prepared mechanochemically did not contain oxides. Due to metallic conductivity of intermetallic compounds  $Cu<sub>3</sub>Si$  and  $Cu<sub>5</sub>Si$ , their presence in the contact masses cannot be directly confirmed using Raman spectroscopy. However, for  $Cu<sub>5</sub>Si$  the Raman spectrum of the oxide film from its surface was presented that differs from the typical Raman spectra of copper oxides  $Cu<sub>2</sub>O$ and CuO, which can indirectly indicate the presence of  $Cu<sub>5</sub>Si$ .

The results of Raman spectroscopy analysis of the coppercontaining phases can be useful for their *in situ* identification during monitoring of the direct synthesis of alkoxysilanes and similar processes.

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#### **Electronic supplementary information**

Microphotographs, Raman spectra, and powder X-ray diffraction patterns for the samples explored. For ESI, see DOI: 10.32931/io2218a.

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